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**Performance and degradation of high Performance MEAs
For High Temperature PBI Fuel Cells**

Thomas Steenberg, Hans Aage Hjuler, Carina Terkelsen, Thorsten Holst and
Hector R. Garcia

Danish Power Systems®



Outline

- Introduction
- Performance and quality control
- Durability
- Challenges!



Who is Danish Power Systems (DPS)?

- Strategy: To develop State of the Art MEAs and establish a manufacturing facility
- Strong relations with Danish and international companies, universities and organizations
- Key player and member of the Danish Partnership for Hydrogen and Fuel Cells



DFS contributions – posters:

Hector R. Garcia



UCLM

Titanium PBI composite membranes for high temperature PEMFC. From a 5 cm² single cell to a 150 cm² short stack

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CARISMA 2012

INTRODUCTION

At present, the most interesting candidates for HT-PEMFC membranes are indeed polybenzimidazole (PBI) and its derivatives (phosphoric acid doped PBI membranes). The PBI based membranes has a relative lower proton conductivity, acid leaching tendency, and inadequate mechanical stability under fuel cell conditions that restrict their life and use in PEMFCs. Further doping of phosphoric acid increases the conductivity but at the expense of mechanical stability. Hence, enhancement of proton conductivity and mechanical strength along with increased resistance toward acid leaching is a challenge to be overcome successfully, and a tailor-made composite with desired properties should alleviate many of these issues.

Recently, novel titanium-based PBI composite membranes have been characterized and tested in a single fuel cell of 5 cm². Because of the good results achieved by these membranes, it was decided to scale-up the fuel cell to a short fuel cell stack of 3 MEAs of 50 cm² each one. It was evaluated the performance at different operation conditions. A target to enable the fuel cell commercialization is increase the life service. Thus, long-term essays were carried out. During this study the leached phosphoric acid was measured and different stop stages were carried out, considering the length of all shut-down stages the study lasted more than 1,700 h. The titanium composite PBI based membranes showed a good stability and are good candidates as electrolytes for high temperature PEMFC systems.

EXPERIMENTAL

Scale-up and evaluation of PBI based HT-PEMFC technology from 5 cm² single fuel cell to 150 cm² fuel cell stack (small full-size fuel cell stack).

Evaluation of fuel cells' performances and life time under different operation conditions.

PBI based electrolyte membranes employed: standard PBI (Poly[2,2'-(m-phenylene)-5,5'-bibenzimidazole]) and composite PBI (filler content: 2 wt. % TiO₂).

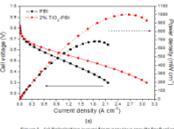


Figure 1. Cell voltages and current densities for fuel cells operating with PBI and 2 wt. % TiO₂ PBI membranes. T = 135 °C, Fuel = H₂, Combuient = O₂, Pressure = 1 atm, for both gases. Membrane area: 50 cm² for 80 wt. % TiO₂, Fuel cell area = 1 cm²/30 cm² for 150 cm² fuel cell stack.



RESULTS AND DISCUSSION

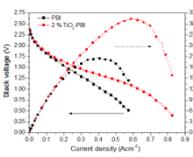


Figure 2. Polarization curves of standard PBI membranes fuel cell stack and composite one. T = 135 °C, Fuel = H₂, Combuient = O₂, Pressure = 1 atm.

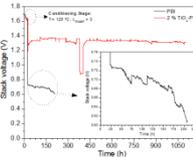


Figure 3. Voltage variation during life test including conditioning stage. Current density = 0.2 A/cm², A_{0.2} = 3, A_{0.6} = 1.7, T = 130 °C, P = 1 atm. (Shut-downs are not included).

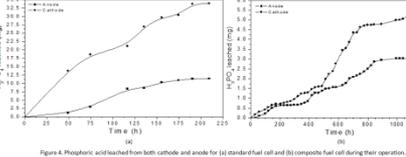


Figure 4. Phosphoric acid leached from both cathodes and anodes for (a) standard fuel cell and (b) composite fuel cell during their operation.

Shut-down number	Operation time (h)	Shut-down length (days)
SFCS	1*	24
	2	166
	3	191
	4	281
CPCS	5	380
	6	174
	7	616
	8	723
	9	882

* Shut-down carried out after conditioning stage

SFCS: Standard Fuel Cell Stack
CPCS: Composite Fuel Cell Stack

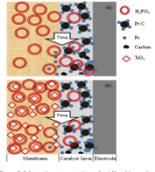


Figure 5. Schematic representations of acid leaching and electrode degradation in standard PBI membrane fuel cell and (b) TiO₂ composite PBI membrane fuel cell. (Figures from: 802 Advances, (2012) 1547-1550).

CONCLUSIONS

Composite Fuel Cell Stack system with TiO₂ composite membranes has shown very significant improvements, if it is compared with the Standard Fuel Cell Stack System. It is a robust system which passes successfully a long life test and which yield a good efficiency. It shows an irreversible cell voltage degradation below 2% during the 1,700 hours of the study (more than 1,100 hours of operation). Furthermore, the leach of acid has been reduced from 2% to 0.6%. All of this is due to the presence of the inorganic reinforcement in the PBI membrane.

ACKNOWLEDGEMENTS

The authors are very grateful with the organization of the III CARISMA that allow us to show the present work.



Danish Power Systems®

HT-PEM - Efficient and renewable energy

Semi-empirical Modelling in a High Temperature Polymer Electrolyte Membrane Fuel Cells Membrane Electrode Assembly (MEA to MEA) Variation

H.R. Garcia, H.A. Hjuler, T. Steenberg, C. Torkelson, T. Holst*

Introduction

Danish Power Systems is working on the manufacture and scale up the main component in HT-PEM fuel systems, the membrane electrode assembly. Semi-empirical modelling is used as an important technique for quality control tests, thus it can reveal the cause of MEA variation and thereby MEA and batch variations can be minimized. The dependence of the kinetic parameters of the model was evaluated by varying the platinum loading (spread) on the electrodes (0.60, 0.77, 1.12 and 1.45 mg Pt/cm²). The ohmic resistance was modelled for a 30, 40 and 80 µm membranes and conductivity measurements were performed. Results show that the membrane resistance is a major contributor to the ohmic losses present in a fuel cell.

Method

The purpose of having one analytical equation for cell voltage prediction is to have an easy understanding of the different losses present in the MEA. The following empirical equation was used assuming:

- Mass transport limitations are negligible
- Species crossover are present in a real fuel cell ($i_{crossover}$)
- Implementation of exchange current density into the logarithmic fitting term ($i_{0,ex}$)

$$E_{cell} = E_{rev} - \ln(i) \left(\frac{RT}{nF} \right) - R \quad \text{Eq. [1]}$$

E_{cell} : Fuel cell output voltage (V)
 E_{rev} : Reversible cell voltage at operation temperature (V)
 n : Tafel slope fitting parameter (mV/dec)
 i : Current density (mA/cm²)
 A_0 : Exchange current density fitting parameter (mA/cm²)
 $i_{crossover}$: Internal crossover current fitting parameter (mA/cm²)
 R : Ohmic resistance fitting parameter (Ω/cm²)

Four MEAs with different Pt loadings were manufactured (see Fig. 1).

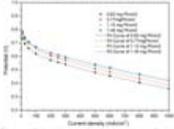


Figure 6. Influence of the platinum loading on the performance. Pt loadings: Carbon/Ink: 0.60(0.33 mg/cm²), 0.70(0.33 mg/cm²), 1.12(0.33 mg/cm²), 1.45(0.33 mg/cm²). (Symbolic). Modelled curves by using Eq. [1]. (Dotted lines). 180 °C.

Table 1. Fitting parameters for the model. Tafel slope n (mV/dec), Ohmic resistance R_{ohm} (Ω/cm²), internal crossover current $i_{crossover}$ (mA/cm²) and exchange current density $i_{0,ex}$ (mA/cm²).

Pt loading (mg/cm ²)	n (mV/dec)	R_{ohm} (Ω/cm ²)	$i_{crossover}$ (mA/cm ²)	$i_{0,ex}$ (mA/cm ²)
0.33	88	0.22	1.8	1.7×10^{-11}
0.77	88	1.18	1.6	8.2×10^{-12}
1.12	89	0.11	5	3.0×10^{-12}
1.45	89	0.19	0.3	1.7×10^{-12}

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Figure 7. Polarization curves for MEAs with different membrane thicknesses (30, 40 and 80 µm), R_{ohm} was fitted by using Eq. [1], 180 °C.

Table 2. Internal crossover current ($i_{crossover}$) and R_{ohm} obtained from Eq. [1], when current voltage (ECV) was measured under no-load operation.

Membrane thickness (µm)	$i_{crossover}$ (mA/cm ²)	R_{ohm} (Ω/cm ²)	$i_{0,ex}$ (mA/cm ²)
30	0.19	0	0.11
40	0.20	0.4	0.19
80	0.22	1.2	0.11

Proton conductivity measurements were carried out in a four probe conductivity cell at 180 °C to verify what is the main issue contribution in a fuel cell. Specific area resistance was obtained.

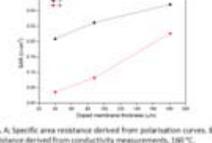


Figure 8. A. Specific area resistance derived from polarization curves. B. Specific area resistance derived from conductivity measurements, 180 °C.

Conclusions

The model for the cell voltage was fitted to the experimental data. The Tafel slope decreases when the Pt loading is increased, which is assumed to be due to an increase in the number of Pt active sites. A linear dependency between open circuit values (OCV) for 30, 40 and 80 µm and the fitted values of the internal crossover current obtained from Eq. [1] was found. Conductivity measurements were carried out to quantify the contribution of the membrane resistance. A linear correlation between membrane thickness and specific area resistance was seen.

Contributions – posters:

Carina Terkelsen



Effect of Humidity on HT-PEM Fuel Cell Manufacture in All Steps From PBI Synthesis to MEA Storage

C. Terkelsen*, H.A. Hjulær, T. Steenberg, A. Nielsen, M. Vajic, Martin Brørholt Sørensen*

Introduction

Water management normally not an issue associated with the operation of high temperature polymer electrolyte membrane (HT-PEM) fuel cells. However, during the manufacturing process humidity plays a large role in the change from the synthesis of polybenzimidazole (PBI) over PBI dissolution to membrane doping, membrane storage and on Membrane Exchange (MEA) storage.

Synthesis

The synthesis of PBI from 3,3'-diaminodiphenyl sulfone (DDS) and isophthaldiamide in a slow growth polymerization and 11 days, very dependent on having a 1:1 stoichiometric ratio of these two monomers. The reaction of water in the DDS is therefore a very important parameter to measure in order to obtain a high molecular weight of the synthesized PBI.

System	Water content (wt%)
DDS	~18%
isophthaldiamide	~11-15%
isophthaldiamide	~11-15%

Table 1. The water content (wt%) of the monomers used.

Figure 1. Water content (wt%) of the monomers used.

By measuring the water content and keeping this constant an 18% PBI can be synthesized. The water content of the PBI product, obtained in the second step (Figure 2).

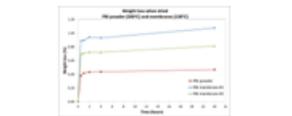


Figure 2. Water content (wt%) of the monomers used.

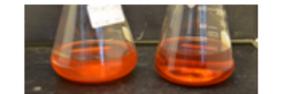


Figure 3. Solution of PBI in DDS (wt) and solution of PBI in DI (wt).

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Mechanical properties

Leaking of gas membranes on the table exposed to the relative humidity of the air influences the doping level of the sample. The membrane also up to a certain extent in the right of the sample. But to perfectly doping of the membrane on the surface decrease in doping level is a natural result of the air in a more dry membrane (Figure 4 and 5).

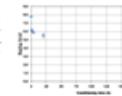


Figure 4. Doping level (wt%) of PBI in DI as a function of doping time.

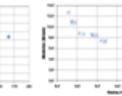


Figure 5. Doping level (wt%) of PBI in DI as a function of doping time.

Figure 6. The doping level (wt%) of PBI in DI as a function of doping time.

Figure 7. The doping level (wt%) of PBI in DI as a function of doping time.

MEA assembly and storage

When assembling the MEA, keeping it in a water solution in the membrane is quite important. If the membrane dries before the MEA is assembled, the MEA will not be able to assemble properly. The MEA will be damaged and the performance will be low.

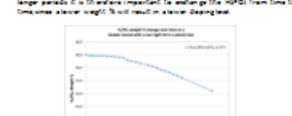


Figure 8. Water content (wt%) of the MEA as a function of time.

Figure 9. The effect of humidity on the MEA performance. The graph shows that the performance of the MEA is significantly lower when the humidity is low compared to when it is high.

Conclusion

Controlling the level of humidity in all the steps from the PBI synthesis to the MEA performance and storage is a difficult task. However, an illustration in this poster is very important in order to reach uniform PBI membrane and MEA quality.



Contributions – posters:

Thorsten Holst



Low Energy X-ray Imaging Used to Quantify the Large-Area Thickness Variation of the Catalyst Loading on Carbon Cloth Based Electrodes for Fuel Cells.

T. Holst*, H.A. Hjuler, T. Steenberg, C. Tørkelsen, H.R. Garcia.

Introduction

Fuel cells based on polymer electrolyte membranes often have microporous carbon cloth electrodes. The necessary catalytic activity is established by coating the carbon cloth surface with a suitable catalyst material, typically nano-sized platinum particles loaded onto a high-surface area carbon structure. From a production point-of-view, it is highly desirable to be able to investigate if the catalyst layer is distributed evenly over the whole electrode area of the fuel cell.

Method

In this study, we introduce an X-ray imaging technique, which can be used to characterize the thickness variation of the Pt-loaded catalyst over a large area of, say, 20 x 20 cm². Low energy X-ray (LEI) imaging is characterized by a high level of gray-scale contrast. The image contrast is mainly provided by the heavy element content of the X-ray. In this case Pt being the heavy element. This makes us a technique particularly suited for revealing even minute changes in the electrode Pt-loading. A spatial image resolution in the sub-millimeter range is easily achieved, which also helps mapping out some coating artifacts.



Figure 3. LEI - The Pt-distribution on a carbon cloth electrode. Right: Reference image of the carbon cloth.

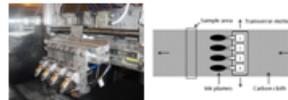


Figure 4. LEI - A spray arrangement for Pt-loading on a carbon cloth electrode. The spray arrangement is used to spray a Pt-solution onto a carbon cloth electrode. The spray arrangement is used to spray a Pt-solution onto a carbon cloth electrode. The spray arrangement is used to spray a Pt-solution onto a carbon cloth electrode.

Results

Image software (ImageJ) was used to analyze the Pt distribution, see Fig. 3 and 4. An unsprayed piece of carbon cloth was used to establish a reference level of the X-ray signal. This zero-loading reference level is seen at the top horizontal axis of Fig. 3 and 4. The average Pt loading was established by weighing the carbon cloth before and after the spraying. The loading levels were then compared to the average change of the X-ray signal. The results are shown in Fig. 5.

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Energy Dispersive X-ray Analysis Used to Quantify the Phosphoric Acid Doping Level in Polybenzimidazole Based Fuel Cells.

T. Holst*, H.A. Hjuler, T. Steenberg, C. Tørkelsen, H.R. Garcia, J.O. Jensen*, L.N. Cleemann*, Q. Li*, N.J. Bjerrum*.

Introduction

In high temperature polymer electrolyte membrane (HT-PEM) fuel cells, the membrane is made of polybenzimidazole (PBI). The necessary proton conductivity of the membrane is ensured by doping it with phosphoric acid (PA). The amount of PA contained in the membrane is described by the doping level (DL), which is the number of PA molecules per repeating unit of PBI. It is important for the HT-PEM fuel cell performance that the doping level is high from the beginning and that it will remain high during the lifetime of the cell. Therefore, it is desirable to have a method, which can quantify the membrane doping level - for example as a post-mortem analysis after a durability test in order to monitor a possible loss of PA overtime.

Method

We have used energy dispersive X-ray spectroscopy (EDS) to establish the amount of PA in the membrane electrode assembly (MEA) structure. Elemental analysis of the membrane's content of phosphorus and oxygen from the PA yields information about the doping level. Samples were prepared by making a cross-section cut through the MEA structure. A spin milling was used to gently do a final polishing of the cross-sectional surface, before the sample entered the scanning electron microscope (SEM). The EDS spectra were acquired from a 10 x 10 μm² area in the center of the 20-50 μm thick membranes and analyzed with the INCA software from Oxford Instruments, see Fig. 6.

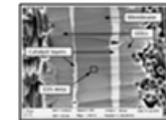


Figure 6. Cross-sectional SEM image of a fuel cell MEA structure. EDS measurements of the PA content were made in the area shown in the range 0.1-1.0 μm.

The membrane's molecular composition is assumed to have the following form:

$$P(\text{M})_x = \frac{DL}{M(\text{PBI})} \cdot M(\text{PBI}) + \frac{DL}{M(\text{PBI})} \cdot M(\text{PA})$$

$$C(\text{O})_y = \frac{DL}{M(\text{PBI})} \cdot M(\text{PBI}) + \frac{DL}{M(\text{PBI})} \cdot M(\text{PA})$$

where the molar masses are: $M(\text{PA}) = M(\text{H}_2\text{PO}_4) = 98.0 \text{ g/mol}$, $M(\text{PBI}) = M(\text{C}_8\text{H}_6\text{N}_2) = 206.2 \text{ g/mol}$. The hydrogen contribution is neglected, since EDS is not sensitive to this element. The difference between the model and the actual measured EDS-values is described by the following root-mean-square error function:

$$f_{\text{error}} = \sqrt{\frac{\sum (y_i - \hat{y}_i)^2 + \sum (x_i - \hat{x}_i)^2}{n}}$$

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where y_i - wt % (model) - wt % (EDS). The doping level is now defined as the value of DL, which minimizes f_{error} . This is equivalent to the traditional least-square method.

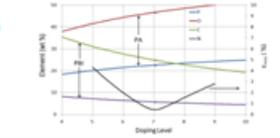


Figure 7. The relative error after the model and the distribution of the four elements as a function of the doping level. An example of the error function f_{error} is also shown in the case where the molar DL = 1.0-1.1. Coverage: 400x10³.

Results

Doping levels in the range 5-8 with an uncertainty of 0.5 were measured. This uncertainty is based on the observation, that the calculated doping levels depend somewhat on the SEM acceleration voltage. In Fig. 8, doping level measurements are compared with durability data for three different MEAs.

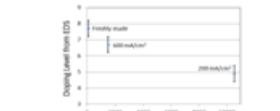


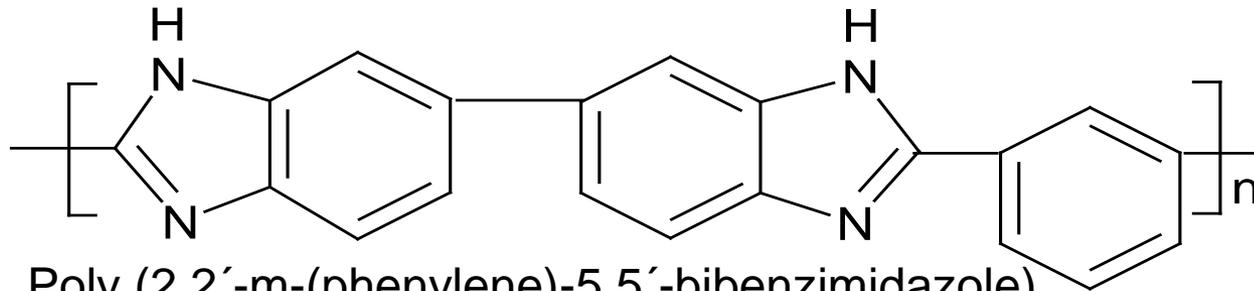
Figure 8. Doping level measurements compared with durability data. DL with error bars and f_{error} are shown in the plot. The values are DL = 7.0, 7.5 and 8.0. The current density used during the durability study.

Conclusion

A method, based on EDS measurements, to quantify the phosphoric acid doping level of HT-PEM fuel cell membranes is suggested. From the EDS measurements of the membrane's elementary constituents, P, O, N, and C, it can be seen that phosphoric acid is lost over time as expected. Even after ~10,000 hours of operation, the doping level is 5, which indicates that the durability tests only in part is a measure of the loss of phosphoric acid in the membrane.



High Temperature PEM FC



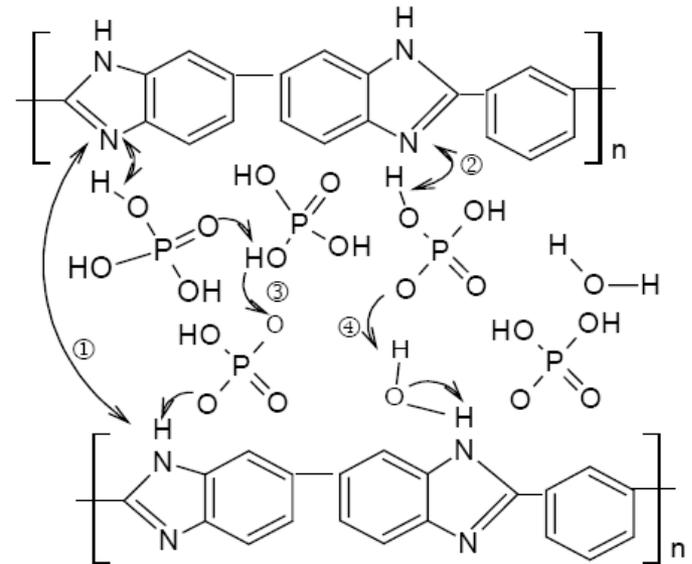
Poly (2,2'-m-(phenylene)-5,5'-bibenzimidazole)

Well-known temperature resistant polymer

$T_g = \sim 430^\circ\text{C}$

When doped with phosphoric acid:

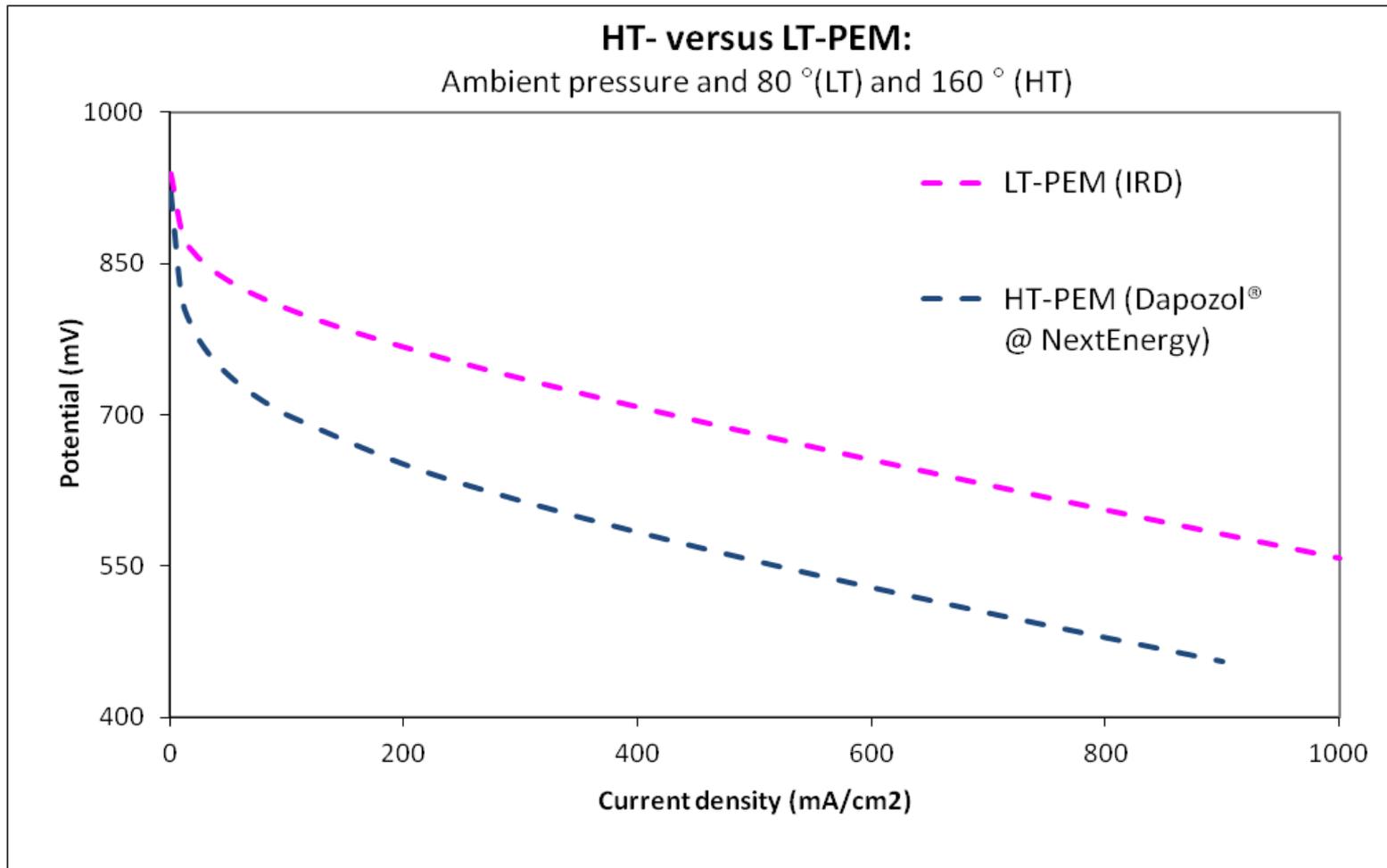
Proton conductor



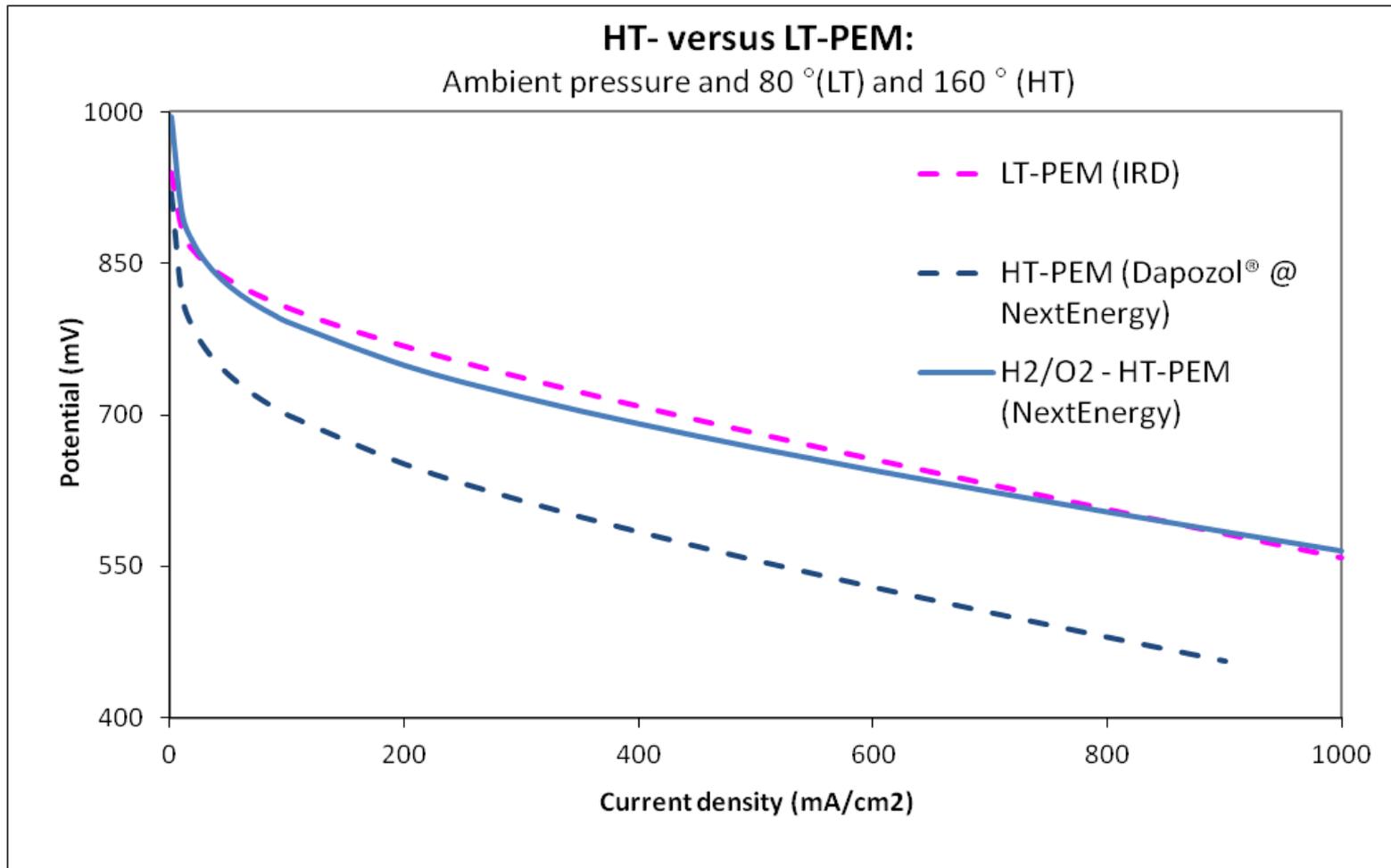
Wainright and Savinell. J. Electrochem. Soc. 142 (1995) L121



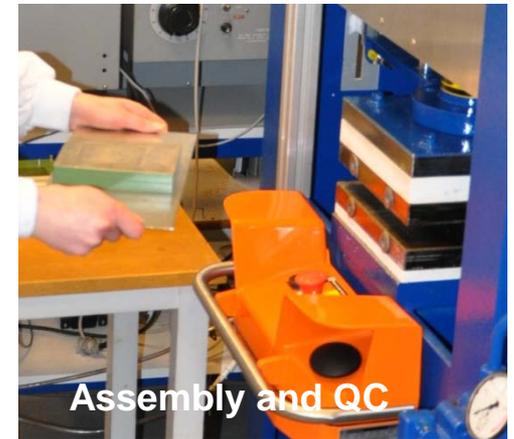
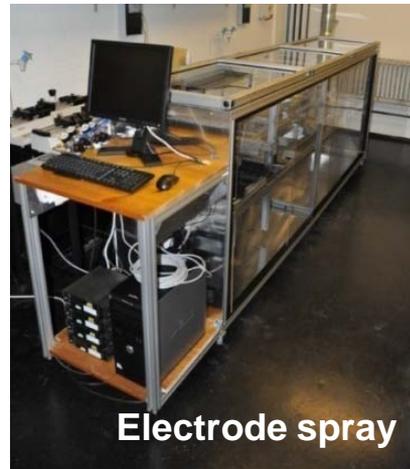
Why HT-PEM?



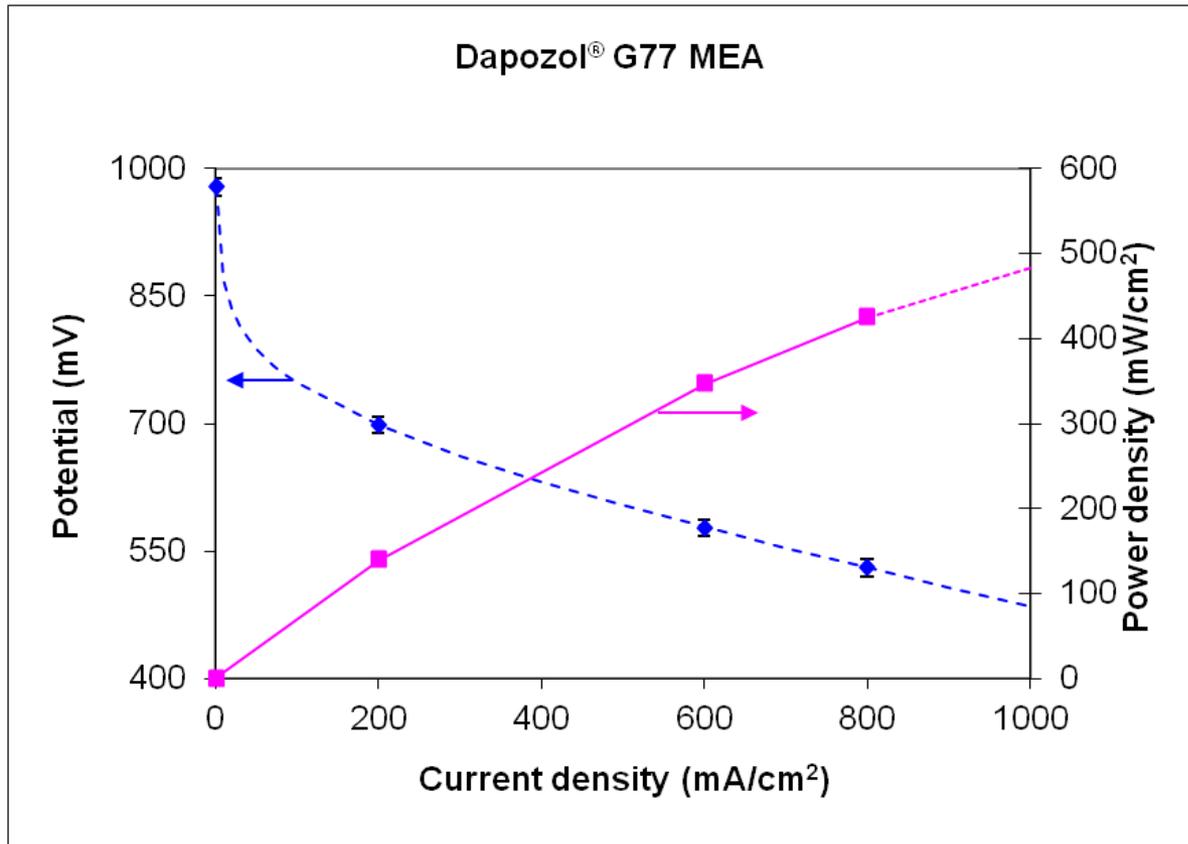
Why HT-PEM?



DPS' capabilities - MEA manufacturing

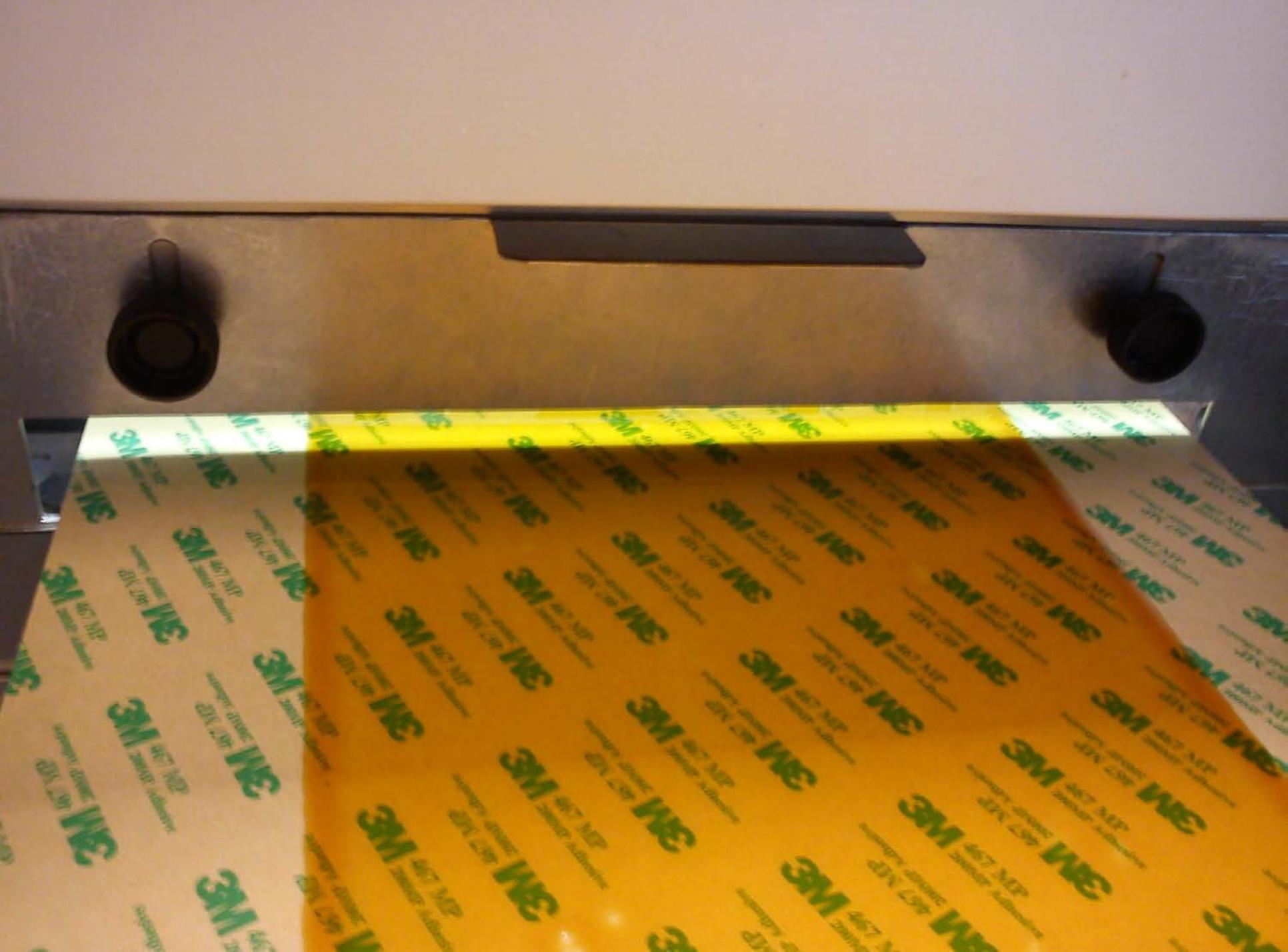


Polarization curve

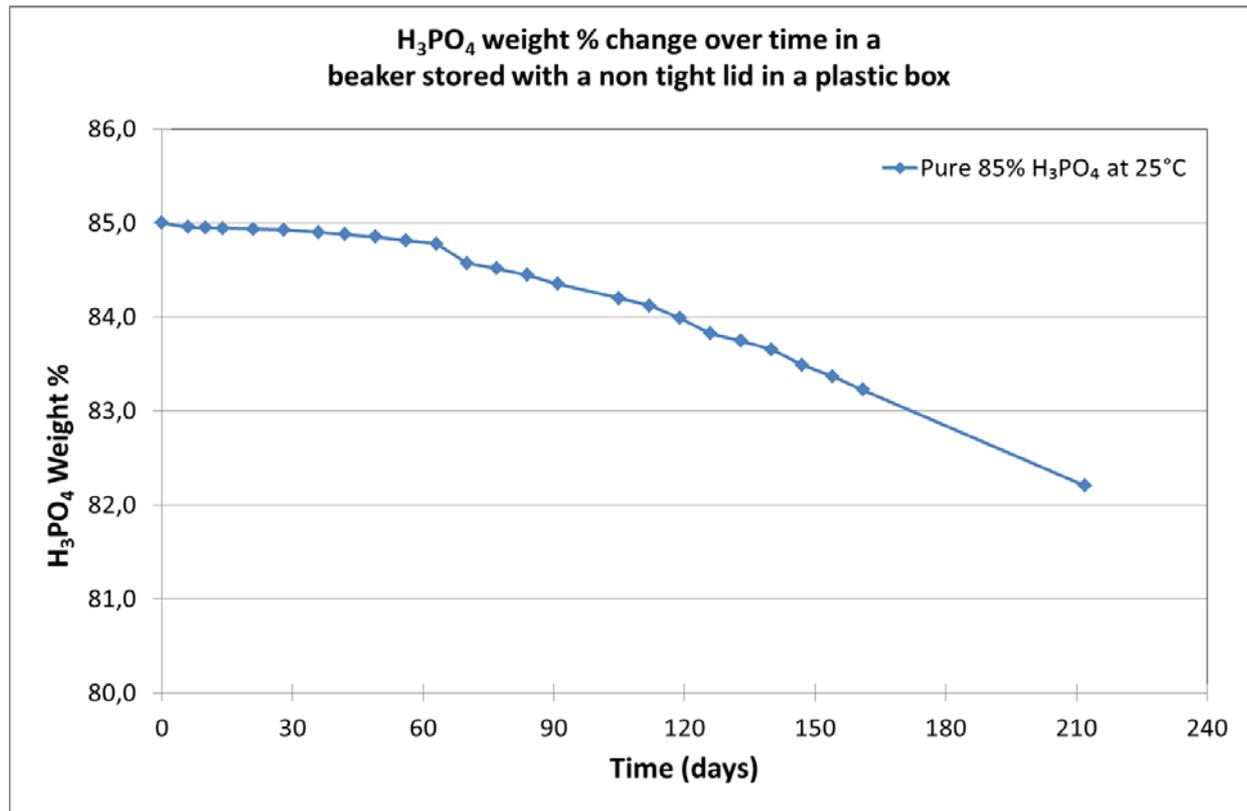


Polarization and a power curve for MEAs tested with hydrogen at 160 °C. Based on 90 identical 50 cm² MEAs. Error bars on the polarization curve indicates standard deviation below 2 %.

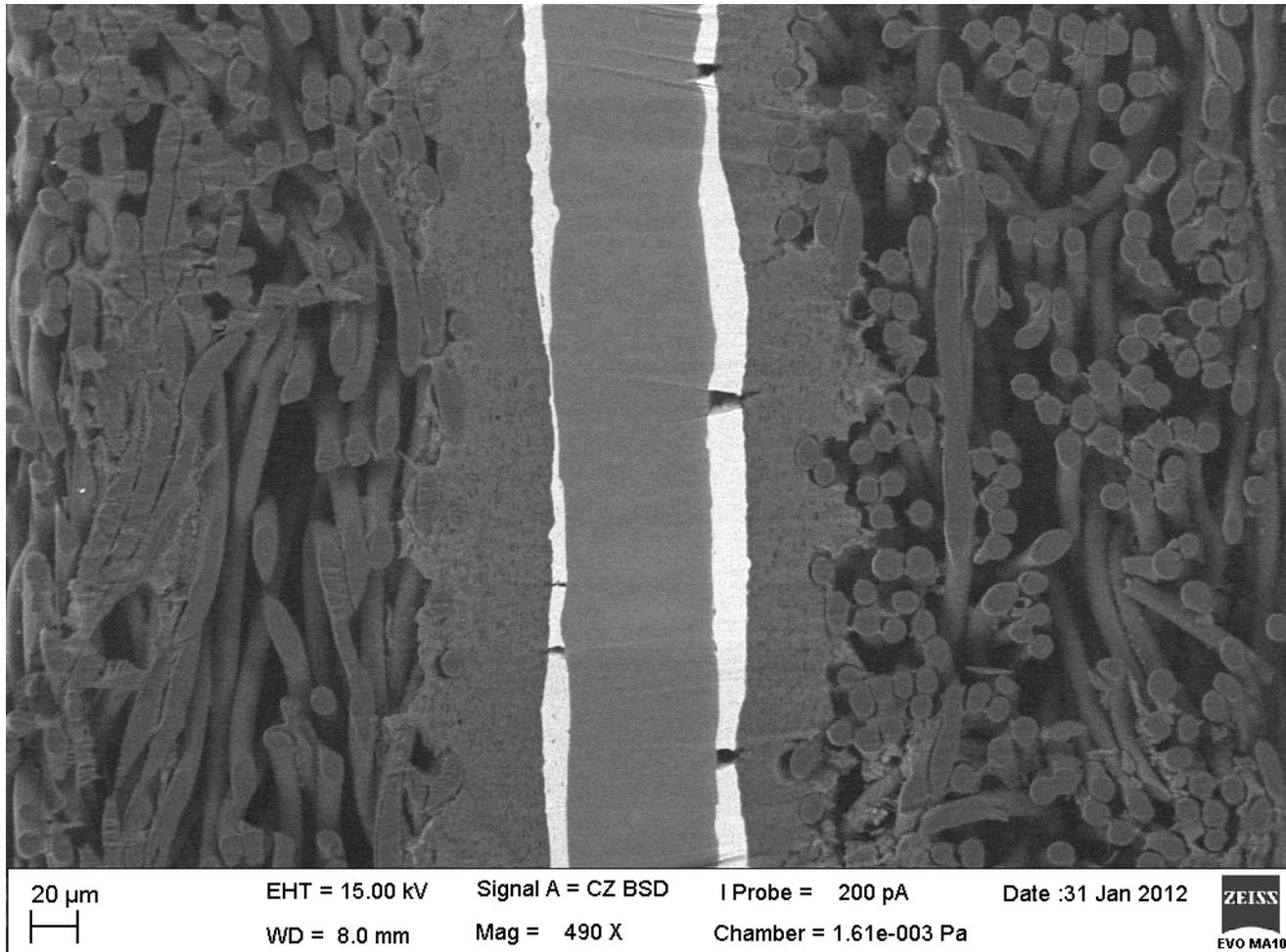




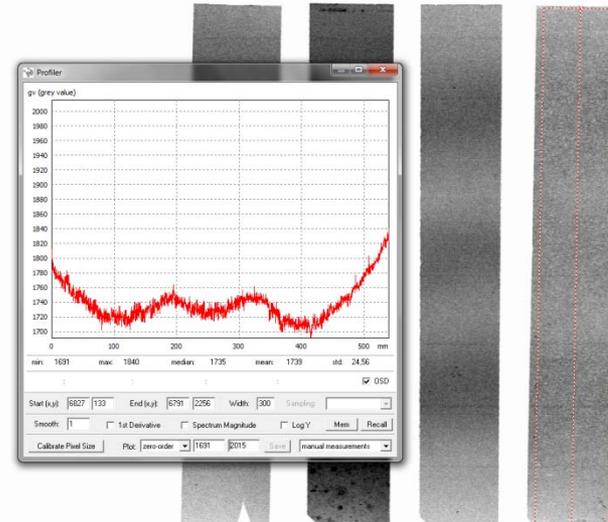
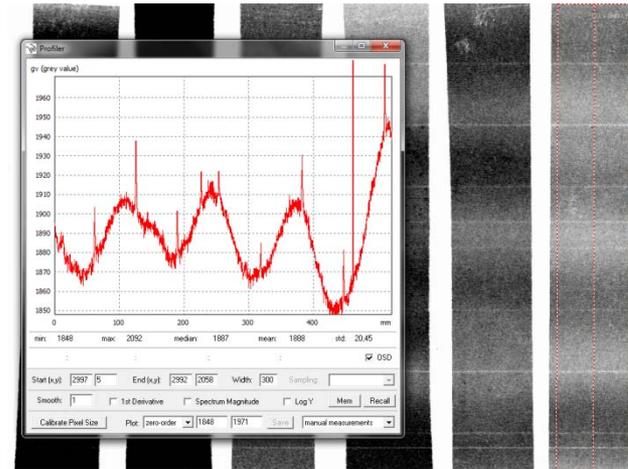
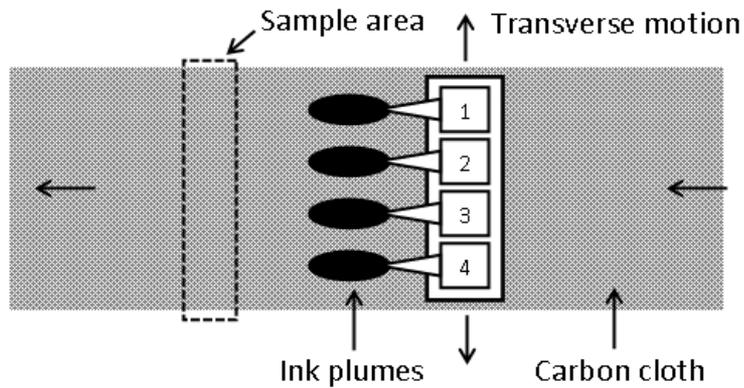
H₃PO₄ and doping:



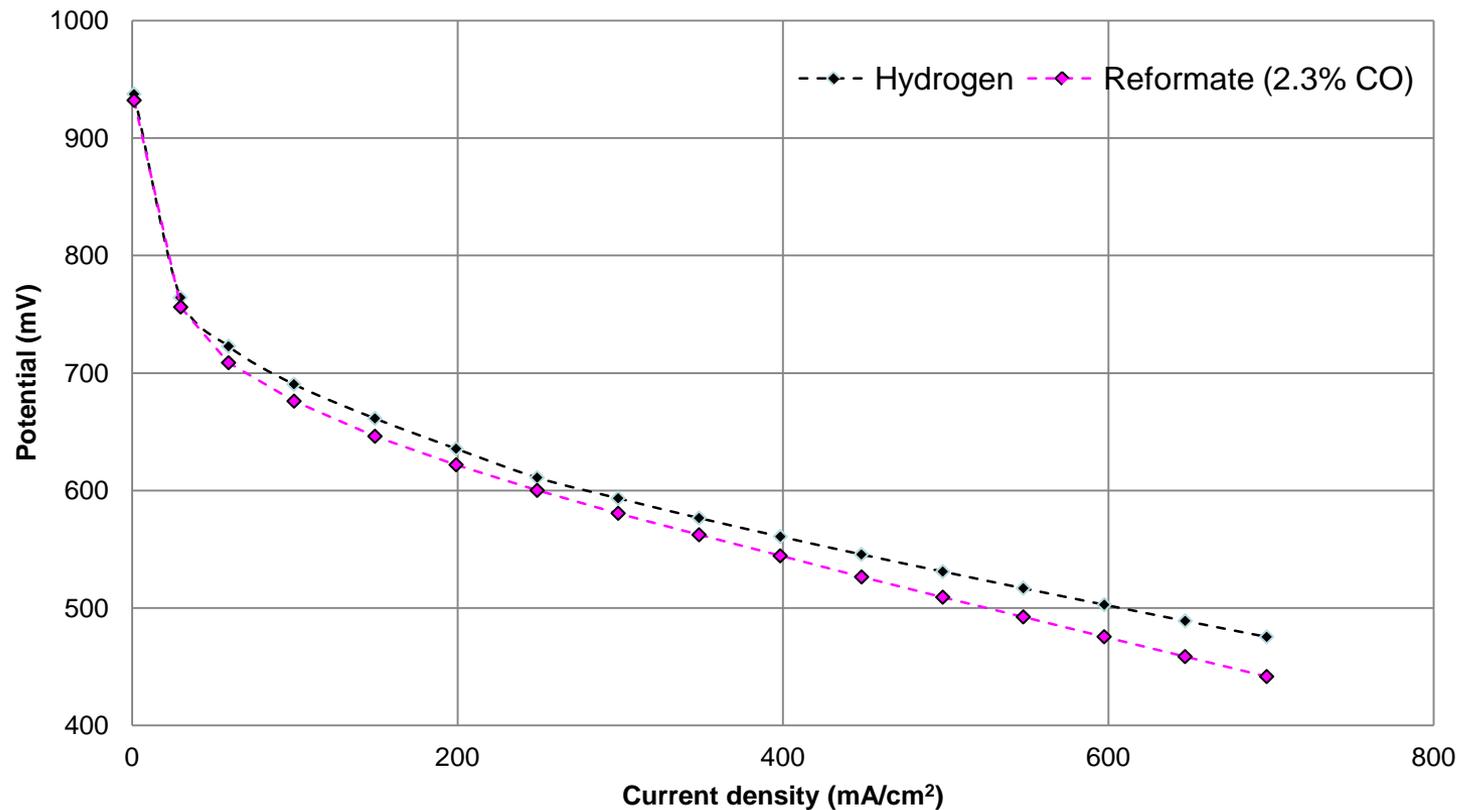
MEA cross sections – ion milled



Electrodes and Pt distribution:



Polarization - hydrogen and reformat

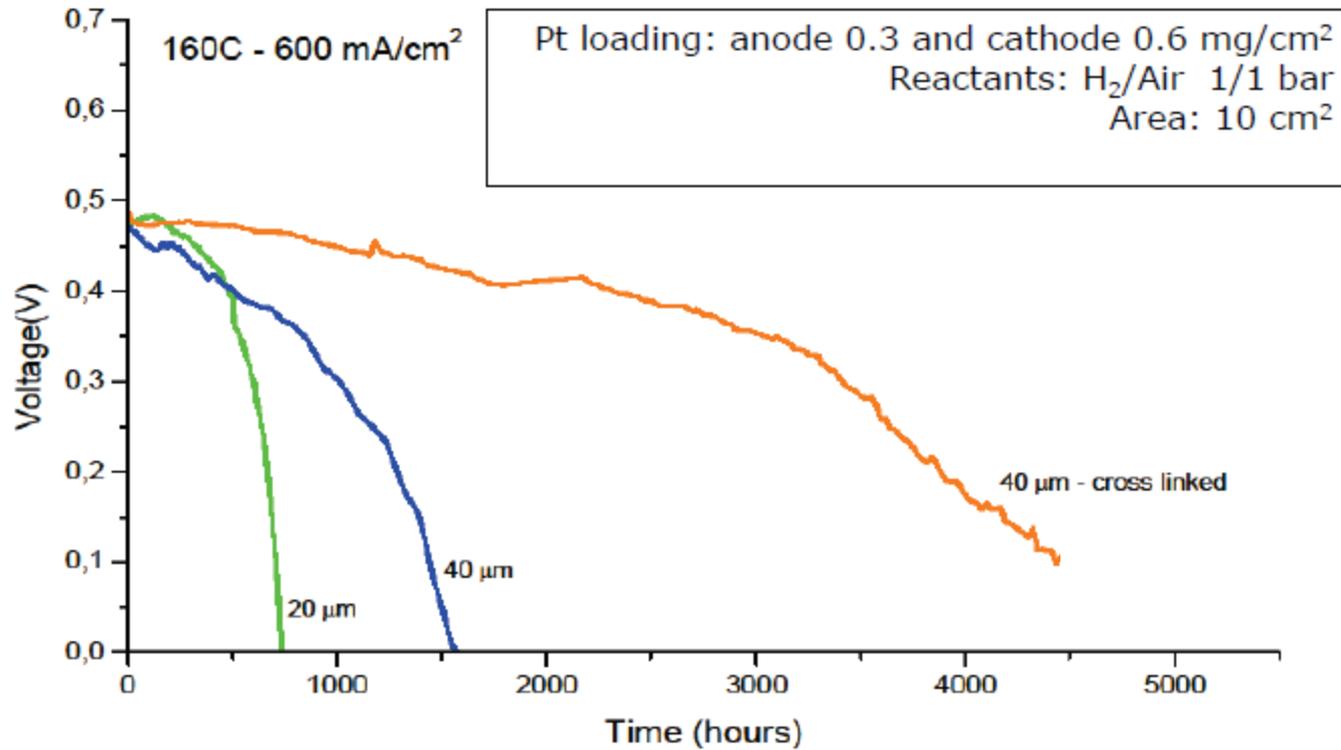


Polarization curve for MEAs tested with hydrogen and reformat at 160 °C.



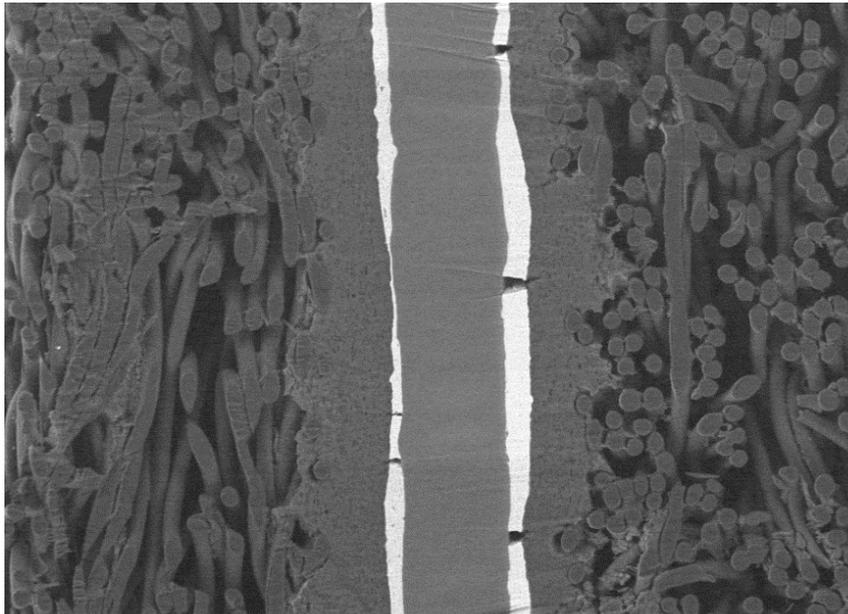
Durability - 600mA/cm²

Durability test – 600 mA/cm²



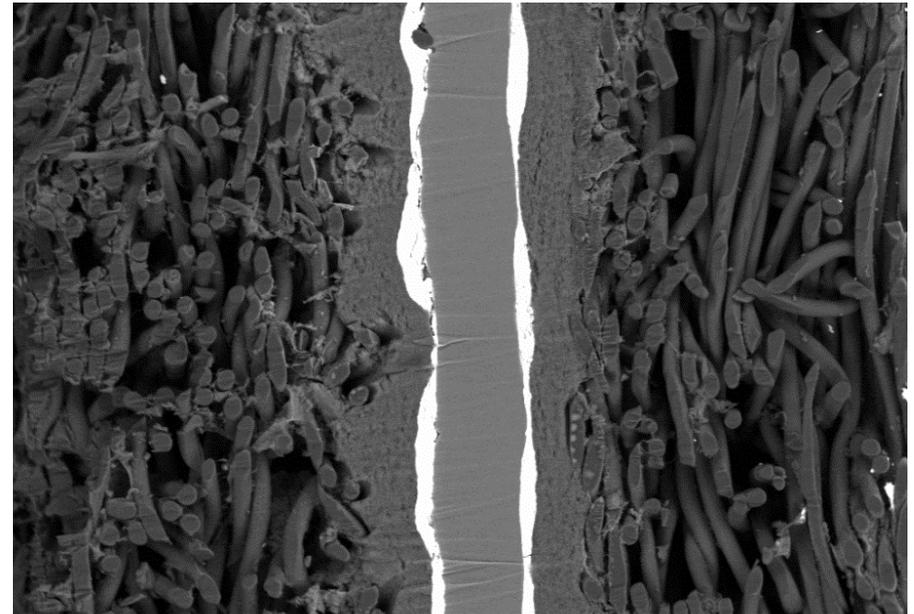
SEM investigations:

Un-used MEA:



20 μ m EHT = 15.00 kV Signal A = CZ BSD I Probe = 200 pA Date :31 Jan 2012
WD = 8.0 mm Mag = 490 X Chamber = 1.61e-003 Pa

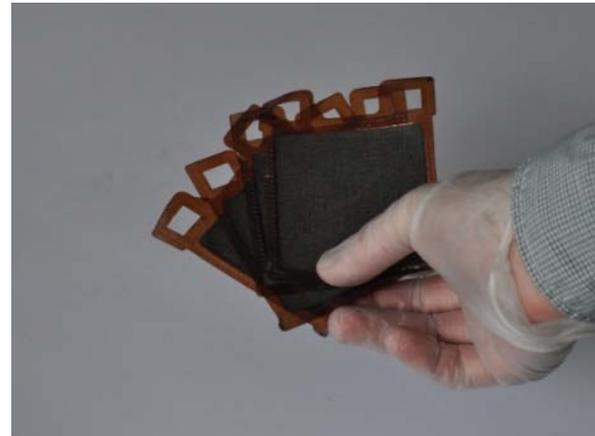
After 4000h operation:



20 μ m EHT = 15.00 kV Signal A = CZ BSD I Probe = 200 pA Date :6 Feb 2012
WD = 11.5 mm Mag = 508 X Chamber = 1.88e-003 Pa

DPS technology status:

- PBI synthesis (uniform Mw)
- PBI membranes (standard and x-linked membranes)
- MEA's (standard and custom sizes)
- Performance and durability



Next step:

- Pilot production in preparation
- Performance improvements
- Durability improvements
- Cost reduction
- It takes time:
 - Demonstration of lifetime
 - Market development



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Thank you!



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